



# H<sub>2</sub>S GAS SENSING BEHAVIOR OF MODIFIED LaCoO<sub>3</sub> THICK FILM A. V. KADU<sup>\*</sup> and S. V. JAGTAP<sup>a</sup>

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# ABSTRACT

Thick films of pure and various concentrations of  $La_{1-x}Sr_xCoO_3$  (x = 0.0, 0.1, 0.2, 0.3 & 0.4) were prepared by solgel technique. Morphological and structural properties of the samples were obtained using the scanning electron microscopy (SEM) and X-ray diffraction techniques respectively. The H<sub>2</sub>S gas sensing properties of these thick films were investigated at different operating temperatures and H<sub>2</sub>S gas concentrations. The surface resistance of thick films decreases when exposed to H<sub>2</sub>S gas. The La<sub>1-x</sub>Sr<sub>x</sub>CoO<sub>3</sub> based sensor with x = 0.3 shows better sensitivity towards H<sub>2</sub>S gas at an operating temperature 210°C. The effect of Sr doping on sensitivity, response time and recovery time of the sensor in the presence of H<sub>2</sub>S and other reducing gases were studied and discussed.

Key words: H<sub>2</sub>S gas sensor, La<sub>1-x</sub>Sr<sub>x</sub>CoO<sub>3</sub>, Sensitivity & Selectivity.

# **INTRODUCTION**

Sensors play an important role in the areas of emissions control, environment protection, public safety, and human health. Much more public concern today than ever before over erious environmental issues is further promoting the development of sensors with both high sensitivity and rapid response<sup>1,2</sup>. In addition to in-depth exploration of gas sensing mechanisms. The use of sensors is basically for measurements of physical quantities and for controlling some systems. Today our environment is polluted by number of gases exhausts from auto and chemical industry. In order to detect measure and control the gases one should know the amount and type of gases present in the ambient. Therefore the need to monitor and control these gases has led to the research and development of very large variety of sensors using different materials and technologies.

 $H_2S$  is a colourless, toxic, flammable and malodorous gas as sources from gasoline, natural gases, city sewage, volcanic gases and hot springs with smells like rotten eggs. It can also be produced from bacterial breakdown of organic matter or wastes produced by human and animal. Other sources are craft paper mills, tanneries and petroleum refineries.  $H_2S$  gas is badly harmful to human body and the environment. Meanwhile the type of oil and natural gas is correlative with the concentration of  $H_2S$ . The oil and natural gases mines can be found depending on the concentration of  $H_2S$ . Therefore, the detection and monitoring of  $H_2S$  are of high importance for both resource exploitation and human health. In the recent

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researchers, a number of semiconductor sensors have been found to be sensitive to  $H_2S$  including  $WO_3$ ,  $In_2O_3$ , ZnO and a few pervoskite type materials<sup>3-7</sup>.

The pervoskite oxides (ABO<sub>3</sub>) were used as gas sensor materials for their stability in thermal and chemical atmospheres. Modifications in microstructure, processing parameters and also concentration of acceptor/donor dopant can vary the temperature coefficient of the resistance and conductivity of ABO<sub>3</sub> oxides. Sensors based on ABO<sub>3</sub>-type complex oxide material, of rare earth elements have an outstanding merit of its high sensitive and selective characteristics. These characteristics can be controlled by selecting suitable A and B atoms or chemically doping A' and B' elements equivalent respectively to A and B into ABO<sub>3</sub> to obtain  $A_xA'_{1-x}B_yB'_{1-y}O_3$  compound<sup>8,9</sup>.

In the recent years, a number of semiconductor sensors have been found to be suitable for  $H_2S$  gas e.g.  $SnO_2$ ,  $WO_3$ ,  $In_2O_3$ ,  $ZnO_2$  and a few perovskite-type materials like NdFeO<sub>3</sub> and NiFeO<sub>4</sub><sup>10-13</sup>. LaCrO<sub>3</sub> based compounds are usually synthesized by the traditional solid state reaction method, urea combustion method<sup>14,15</sup>. Sol–gel approach, as a classical method, has a lot of inherent merits with better homogeneity, higher purity, and lower temperature of preparation. The aim of present work is to prepare nanocrystalline  $La_{1-x}Sr_xCoO_3$  thick films by sol-gel route and to investigate their sensing properties for  $H_2S$  gas.

#### EXPERIMENTAL

# Material synthesis

The La<sub>1-x</sub>Sr<sub>x</sub>CoO<sub>3</sub> (x = 0.0, 0.1, 0.2, 0.3 & 0.4) were synthesized by sol-gel method using ethylene glycol as a solvent. All reagents were of analytical grade were used. Firstly, the analytically pure grade La  $(NO_3)_3$ ·6H<sub>2</sub>O, Sr  $(NO_3)_2$ .6H<sub>2</sub>O, Co  $(NO_3)_3$ ·9H<sub>2</sub>O and citrate acid were weighed and dissolved in ion-free water at 80°C for 3 h. Then ethylene glycol was added under constant stirring to obtain a homogeneous and stable sol. The solution was further heated in pressure vessel at about 130°C for 12 h. The material was then heated in a furnace at 350°C for 3 h. The powder was then calcined at 800°C for 6 h in order to improve the crystallinity of materials.

### **Characterization of samples**

The synthesized samples were characterized by powder XRD using a Siemens D 5000 diffractometer. The XRD data were recorded by using Cu K $\alpha$  radiation (1.5406 Å). The intensity data were collected over a 2  $\theta$  range of 10-70°. The average crystallite size of the samples was estimated with the help of Scherrer equation using the diffraction intensity of all prominent lines. The microstructure of the materials was evaluated by scanning electron microscopy (SEM).

# Fabrication and analysis of gas sensors

The prepared powders were mixed with PVA (polyvinyl alcohol) to form paste, and then the paste was coated onto an  $Al_2O_3$  tube on which two platinum wires had been installed at each end. A small Ni–Cr alloy coil was placed through the tube as a heater, which provided operating temperature at 50-350°C. The temperature was controlled by adjusting the heating power. Gas response is defined as the ratio of change in resistance of the sample on exposure to a test gas to the resistance in the presence of air.

$$Sensitivity(S) = \left| \frac{R_a - R_g}{R_a} \right| = \frac{\Delta R}{R_a}, \qquad \dots (1)$$

where Ra is the resistance in air, Rg the resistance in a sample gas and  $\Delta R$  the change in resistance. The response of the sensor for H<sub>2</sub>S was tested in the presence of other gases so that the selectivity can be determined.

# **RESULTS AND DISCUSSION**

## X-ray diffraction study

The XRD patterns of the compositions LaCoO<sub>3</sub> and La<sub>0.7</sub>Sr<sub>0.3</sub>CoO<sub>3</sub> prepared by sol-gel method calcined at 800°C are presented in Fig. 1. Fig. 1(a), shows the XRD pattern, which is in good agreement with XRD results previously reported in the literature for LaCoO<sub>3</sub><sup>16</sup>. The diffraction data is good agreement with JCPD card of LaCoO<sub>3</sub> (JCPDS No. 24-1016). All the diffraction peaks of the phases are indexed as perovskite-type with tetragonal structure. Fig. 1(b), shows the XRD pattern of La<sub>0.7</sub>Sr<sub>0.3</sub>CoO<sub>3</sub> nanomaterial. Extremely broad reflections are observed indicating nanosized particle nature of the material obtained. The mean crystallite sizes (D) of La<sub>0.7</sub>Sr<sub>0.3</sub>CoO<sub>3</sub> powder was deduced from half height width of XRD peaks based on the Scherer's equation,  $D = 0.9\lambda/\beta cos\theta$ , where D is the average size of the particles,  $\lambda$  is wavelength of X-ray radiation,  $\beta$  the full width at half maximum of the diffracted peak and  $\theta$  is the angle of diffraction<sup>17</sup>. The average particle size of the nanocrystalline La<sub>0.7</sub>Sr<sub>0.3</sub>CoO<sub>3</sub> according to the scherrer formula was in the range of 30–35 nm.



Fig. 1: XRD spectra of (A) LaCoO<sub>3</sub> (B) La<sub>0.7</sub>Sr<sub>0.3</sub>CoO<sub>3</sub> calcined at 800°C

#### Sem analysis

Fig. 2(a), undoped LaCoO<sub>3</sub> sample shows randomly distributed grains with smaller size and large agglomeration of particles are observed. Fig. 2(b), shows the microstructure of a most sensitive sample  $La_{0.7}Sr_{0.3}CoO_3$  consisting of giant grains of Sr-species associated with smaller grains of LaCoO<sub>3</sub>. This shows more porosity, giving largest effective surface area. It gives clear evidence for a porous structure, which together with the small particle size points at the films being excellent for gas sensing applications.



Fig. 2: SEM micrograph of samples (a) LaCoO<sub>3</sub> and (b) La<sub>0.7</sub>Sr<sub>0.3</sub>CoO<sub>3</sub>

#### Gas sensing characteristics

Electrical characterization of samples  $La_{1-x}Sr_xCoO_3$  (x = 0.1, 0.2, 0.3 & 0.4) synthesized by sol-gel method and calcinied at 800°C has been carried out in a gas test chamber. The resistance of the films decreased upon exposure to H<sub>2</sub>S gas. Fig. 3 shows gas sensing response of H<sub>2</sub>S gas, at different operating temperatures (50-400°C) for thick films of pure LaCoO<sub>3</sub>. The enhanced sensitivity with increasing the temperature can be attributed to two facts. On one hand, the thermal energy obtained is high enough to overcome the activation energy barrier of the surface reaction; on the other hand, the adsorbed oxygen species convert from O<sub>2</sub> – to O– at elevated temperatures, and thus an increase in electron concentration results from the sensing reaction of a test gas and O–<sup>18</sup>.

![](_page_3_Figure_3.jpeg)

Fig. 3: H<sub>2</sub>S gas response vs operating temperatures for LaCoO<sub>3</sub> calcined at 800°C

Thus, the pure LaCoO<sub>3</sub> shows response towards  $H_2S$  gas at high operating temperature 300°C, which is rather high and practically inconvenient in view of commercial standards. Therefore, efforts were made to modify the LaCoO<sub>3</sub> based sensors by doping with Sr, so as to be operated at lower operating temperatures with high sensitivity and selectivity. Fig. 4 shows gas sensing response of  $H_2S$  gas, for thick films of pure LaCoO<sub>3</sub> doped with different concentrations of Sr. From the figure it is clear that the response values of every doped sample apparently increased with increasing the operating temperature.

![](_page_3_Figure_6.jpeg)

Fig. 4: H<sub>2</sub>S gas response Vs operating temperatures for (A) La<sub>0.9</sub>Sr<sub>0.1</sub>CoO<sub>3</sub>, (B) La<sub>0.8</sub>Sr<sub>0.2</sub>CoO<sub>3</sub>, (C) La<sub>0.7</sub>Sr<sub>0.3</sub>CoO<sub>3</sub> & (D) La<sub>0.6</sub>Sr<sub>0.4</sub>CoO<sub>3</sub> calcined at 800°C

In this figure, it is observed that the  $La_{0.7}Sr_{0.3}CoO_3$  thick film had the largest sensing response at an operating temperature 210°C. It is indicating that Sr doping can greatly improve the sensitivity of LaCoO<sub>3</sub>

based sensor towards  $H_2S$  gas. We know that the semiconductor oxide gas sensors can detect different gases by changing the conductivity of their surface due to the reaction of the reducing gases with adsorbed oxygen. Sr doping in LaCoO<sub>3</sub> promotes the deviation from stoichiometry and enhances the surface defect. A large surface defect concentration creates more active surface states for adsorbed oxygen<sup>19,20</sup>.

Selectivity is the ability that a gas sensor to distinguishes between different kinds of gases. Fig. 5 shows the cross sensitivity of  $La_{0.7}Sr_{0.3}CoO_3$  for NH<sub>3</sub>, LPG and H<sub>2</sub> gases as a function of operating temperature. It is evident from the figure that the  $La_{0.7}Sr_{0.3}CoO_3$  sensor was highly selectivity to H<sub>2</sub>S gas against NH<sub>3</sub>, LPG and H<sub>2</sub> gases. The sensor shows high degree of selectivity towards H<sub>2</sub>S gas than other reducing gases.

Fig. 6 depicts the variation of sensitivity of  $La_{0.7}Sr_{0.3}CoO_3$  sample with H<sub>2</sub>S gas concentrations at 210°C temperature. It is clear from the figure that the gas response goes on increasing linearly with gas concentration up to 200 ppm and saturated beyond it. The rate of increase in gas response was relatively larger up to 200 ppm. Thus, the maximum sensitivity was obtained at higher concentration of H<sub>2</sub>S gas, i.e., 200 ppm.

![](_page_4_Figure_4.jpeg)

Fig. 5: Cross sensitivity vs operating temperatures for different reducing gases for La<sub>0.7</sub>Sr<sub>0.3</sub>CoO<sub>3</sub>

![](_page_4_Figure_6.jpeg)

Fig. 6: Gas Response of La<sub>0.7</sub>Sr<sub>0.3</sub>CoO<sub>3</sub> to H<sub>2</sub>S gas of different concentration at an operating temperature 10°C

The response and recovery times of  $La_{0.7}Sr_{0.3}CoO_3$  sample is represented in Fig. 7. The response was quick (~ 40 s) to 200 ppm of H<sub>2</sub>S while the recovery was moderate (~ 80 s). The quick response may be due to faster oxidation of gas.

![](_page_5_Figure_1.jpeg)

Fig. 7: The response and recovery characteristics to 200 ppm H<sub>2</sub>S gas of La<sub>0.7</sub>Sr<sub>0.3</sub>CoO<sub>3</sub>

## CONCLUSION

Nanocrystalline  $La_{0.7}Sr_{0.3}CoO_3$  has been synthesized by a convenient sol-gel route. The XRD pattern of  $La_{0.7}Sr_{0.3}CoO_3$  shows perovskite-type with tetragonal structure. The results revealed that the particle size is in the range of 30–35 nm for  $La_{0.7}Sr_{0.3}CoO_3$  with good crystallinity. From the results obtained, pure  $LaCoO_3$  showed low response to  $H_2S$  gas. Sr doped  $LaCoO_3$  thick films were found to be high sensitive for  $H_2S$  gas. Among all other synthesized samples  $La_{0.7}Sr_{0.3}CrO_3$  thick film was found to be optimum and showed highest response to  $H_2S$  gas at 210°C. The sensor is very promising for  $H_2S$  detection in the range 200 ppm with a response time in second range.

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